

1

Nanotechnology in Electronics, Materials Properties, and Devices: State of the Art and Future Challenges

P. M. Visakh¹ and Raneesh Balakrishnan²

¹Faculty of Electronic Engineering, Department of Physical Electronics, TUSUR University, Tomsk 634050, Russia

²Department of Physics, Catholicate College, Pathanamthitta, Kerala, 689645, India

1.1 Graphene-based Nanoelectronic Biosensors

Graphene is the thinnest, strongest material, with one-particle thickness, and is 200 times stronger than steel. Graphene exhibits outstanding electrical and thermal conductivity with remarkable light absorption capability. Graphene is a material that could change the world in the true sense, with its infinite potential for integration practically in any industry. Graphene has diverse forms and can be combined with other materials to form various materials of extraordinary characteristics. Researchers are very keen to explore and patent graphene and gain expertise about its various properties and potential applications, mainly in cells, transistors, computer chips, supercapacitors, energy production, etc.

The common graphene derivatives are pristine graphene, reduced graphene oxide, graphene quantum dots (GQDs), and polycrystalline graphene, which are extensively used in biosensing applications. Pristine graphene consists of single crystalline grains that have flawless lattices made up of defect-free hexagonal ring networks of sp^2 hybridized carbons atoms [1]. Graphene oxide also mainly consists of covalently bonded sp^2 -hybridized carbon atoms but has some defects due to the following reasons: disrupted sp^3 -hybridized carbon atoms and the oxygen functional groups like carboxyl group located at the edges and the basal plane are saturated with hydroxyl and epoxy groups [2]. The presence of these groups makes graphene oxide hydrophilic depending on their oxidation state [3].

Nanoelectronic devices are made up of nanomaterials such as nanowires, graphene, carbon nanotubes (CNTs), and transition metal dichalcogenides. These nanomaterials must possess extraordinary characteristics such as high surface-area-to-volume ratio, low power consumption, high charge mobility, and excellent compatibility to fit in modern electronic devices. As nanomaterials display unusual properties that are not obtainable at the micro scale, nanoelectronics holds

significant importance. Graphene is a recently discovered nanomaterial known for its outstanding mechanical, electrical, and optical properties [4–6]. To develop an efficient graphene-based electrochemical biosensor, the graphene oxide nanosheets were decorated with Au NPs by aryldiazonium bridges fabricated on glassy carbon electrode (GCE). Then, the glucose oxidase enzyme was immobilized on the graphene oxide–Au NP nanocomposites. The biosensor showed a high sensitivity of about $42 \mu\text{A mM}^{-1} \text{cm}^{-2}$ with a wide detection range of 0.3–20 mM. This glucose biosensor showed excellent selectivity for glucose detection [7].

N-doped graphene–chitosan–glucose oxidase biosensors showed an excellent sensitivity of $226.24 \mu\text{A mM}^{-1} \text{M}^{-2} 120$ due to high porosity of graphene and high conductivity of graphene sheets, which facilitate enhanced electron transfer. The CuO nanoparticle (NP) composites with graphene oxide have shown a high sensitivity of $262.52 \mu\text{A mM}^{-1} \text{cm}^{-2}$ with an LOD of $0.69 \mu\text{M}$. This biosensor was also used for human serum detection and showed a sensitivity of $285.38 \mu\text{A mM}^{-1} \text{cm}^{-2}$ [8]. Similarly, biosensors based on Pt–NiO–reduced graphene oxide nanocomposites showed a sensitivity of $832.95 \mu\text{A mM}^{-1} \text{cm}^{-2}$ and an LOD of $2.67 \mu\text{M}$. This high performance was attributed to porosity of Pt–NiO–reduced graphene oxide nanocomposites [9]. Moreover, the Cu–Co–reduced graphene oxide nanocomposites were electrochemically deposited over the pencil graphite electrode and showed a high sensitivity of $240 \mu\text{A mM}^{-1} \text{cm}^{-2}$ and an LOD of $0.15 \mu\text{M}$ [10].

1.2 Zinc Oxide Piezoelectric Nanogenerators for Low-frequency Applications

Nanoparticles can be fabricated through solid-phase, liquid-phase, and gas-phase methods. Chemical bath deposition [11], green chemistry synthesis [12], and wet chemical methods [13] are other recently used techniques for the synthesis of ZnO NPs. Vertical ZnO nanorods were grown hydrothermally on a gold-coated PTFE (polytetrafluoroethylene). Spin coating was used to deposit 0.005 M zinc acetate dehydrated in ethanol on the surface of gold-coated PTFE film. Then, the substrate was heated at 90°C for 30 minutes. The seed layer was uniformly deposited in a two-step process of deposition and decomposition to cover the substrate. ZnO nanorods were hydrothermally grown on the seed layer. Zinc nitrate hexahydrate and hexamethylenetetramine (HMT) were used to hold the seed layer at 90°C for three hours [14].

Masuda et al. [15] demonstrated a simple liquid-phase synthesis method to control the morphology and crystal size of ZnO. ZnO nanowires of 100 nm long and about 50 nm wide were successfully synthesized at 50°C . They also found that nanowires had no branches without aggregations. Several techniques have been explored to synthesize and develop both direct and alternating current zinc oxide piezoelectric nanogenerators [16]. Controlled precipitation is typically used to produce ZnO NPs in large quantities with reproducible properties for use in industrial products. In this process, zinc salt solution is spontaneously reduced by a reducing agent.

Breaking down of agglomerations is an important process in this method. Therefore, temperature, pH, type of aqueous media, time of precipitation, and the raw materials are most important parameters in the precipitation of ZnO NPs, which have impact on the size and final characteristics of the nanoparticles [17]. Aladpoosh et al. [18] used the ZnO-controlled precipitation method, a green synthesis method using natural plant source, namely, Keliab and zinc acetate. Wurtzite structure and nanorod shapes result from in situ synthesis of ZnO on the cellulosic chains of cotton using the green synthesis method. The principle of nanogenerators was first used by Wang's research group in 2006 [19]. They used a conductive tip of an atomic force microscope (AFM) to deflect the aligned semiconductor ZnO nanowires and successfully obtained a measurable piezoelectric voltage output of around 10 mV. They grew ZnO nanorods on a sapphire substrate using a vapor–liquid solid method, which caused the gold particles to remain on the top of each rod after their growth.

A Schottky contact was formed between the conductive AFM tip and the semiconductor ZnO nanowires, which is a key factor in current generation by piezoelectric nanogenerators. The stretched side of the rod with a positive potential produced a reverse bias with the Schottky junction, leading to no current to screen the polarization. When the contact between the tip and the compressed, negatively polarized side of the rod was disconnected, the junction was forward biased, and the current was able to flow and screen the polarization. Monitoring the open-circuit voltage reflects an uncertainty in the measurement of this type of device [20]. Zinc oxide is an environmentally friendly multifunctional semiconductor possessing outstanding piezoelectric effect, piezoelectric coupling coefficient, and thermal and mechanical stability at room temperature. Owing to its non-centrosymmetric hexagonal wurtzite crystal structure and polar Zn—O bonds, ZnO has unique piezoelectric properties as the core of studies in the ceramic industry [21, 22]. For the first time in 2009, Xi and coworkers [23] fabricated a ZnO-nanotube-based piezoelectric nanogenerator. They used a solution chemical method to synthesize hexagonal ZnO nanotube arrays at temperatures below 100 °C.

ZnO-based nanogenerators can be incorporated into various substrates. In 2012, Khan et al. [24] grew ZnO nanorods on a silver-coated cotton fabric using a low-temperature, aqueous chemical growth method. An external force was applied on ZnO nanorods via an AFM tip, where the flow of charges was taken into account and a Schottky barrier formed between the electrode and the nanorods.

1.3 Investigation of the Hot Carrier–induced Degradation in Nanoscale Junctionless MOSFETs: A Reliability-based Analysis

In the past decades, reliability of MOSFET devices was a major concern in the electronics industry: the aging phenomenon degraded the device operation and lifetime severely. Extensive research has been carried out to identify the causes of aging and to seek viable solutions. Channel and substrate hot carrier injection

(HCI and SCI), channel hot electron/hole (CHE/H), drain avalanche injection, and radiation were found to be the main degradation mechanisms [25–31]. Hot carrier-induced degradation is of prime interest for designers as the process aggravates with downscaling. Degradation is essentially due to the enhancement of the electric field with dimension reduction, in the lucky electron model representation, providing the carriers enough energy to be injected through the oxide layer. These carriers are most likely to be captured by the existing oxide traps or to break the silicon bonds and thus giving rise to a localized and nonuniform pile-up of interface states near the drain–channel junction [32]. A energy-driven paradigm proposed by Rauch et al. suggests that the electron–electron scattering mechanism, governed by energy rather than the electric field, is responsible for aggravation of hot carrier degradation in downscaled devices [33].

The concept of junctionless transistor was proposed by Lee et al. [34] as a simple way to overcome process limitations. Indeed, as downscaling of channel lengths trends toward less than 10 nm, controlling process-induced variability becomes imperative to conserve the properties of transistors. At these scales, inherent scattering and diffusion of shallow-extension impurities into ultra-short channels pose a serious problem. As high doping concentration gradient is primordial, the extension formation with the actual process techniques implies additional complex and costly operations [35]. Contrary to IM MOSFETs, measures highlighted an enhancement in the effective mobility with the increasing gate bias [36]. The enhancement is so significant that the effective mobility exceeds the corresponding bulk mobility especially in the accumulation regime [37]. Surface mobility also significantly improved in the accumulation layer of heavily doped devices, which indicates a higher current in the accumulation regime of the junctionless field-effect transistor (JLFET). This effect is attributed to the impact of the accumulation layer on screening of ionized impurities, yielding a reduction of Coulomb scattering [38]. In heavily doped devices, measurements highlight a correlation between the carrier density and the Coulomb scattering limitation, where a significant improvement in the effective mobility with respect to the carrier density is observed [39, 40]. Also, it was demonstrated that enhanced electrostatic controllability provided by multigates and thinner equivalent oxide thickness (EOT) yielded higher carrier density and improved mobility.

1.4 Study of Electrostatic and Dispersion Forces in Nanoelectromechanical Systems (NEMS)

Due to the outstanding electrical and mechanical properties of nanostructures, researchers are interested in novel generations of ultrasmall devices such as micro- and nanoelectromechanical systems (MEMS and NEMS, respectively), which are used in switches, actuators, sensors, AFMs, mass and force detectors, etc. New effects appearing due to the scale alteration in geometry introduce several additional terms into the theory of nanoscale structures, while their influence on other scales is negligible. One of these effects is dispersion forces. Dispersion

forces or quantum interactions, including the Casimir and van der Waals (vdW) regimes, play a key role in the behavior of nanodevices [41–43], related to the structure dimensions/configurations and the gap between electrically actuated electrodes [44]. The total electromagnetic Casimir energy is the sum of energies of the Dirichlet and Neumann modes. However, in large separations, the Dirichlet mode is dominant, while the Neumann mode is negligible [45]. It should be noted that the large separation assumption is valid for most real nano-tweezers since the gap between the arms is relatively larger than the radius of the arms ($D \gg R$). Typically, for ($D < 100R$), the contribution of the Neumann mode is negligible compared to the Dirichlet mode, and the asymptotic approximation provides an excellent estimation of the Casimir force [46]. In this section, practical mathematical models by assuming the large separation approximation (Dirichlet mode) are developed.

1.5 Nanomaterials for Wearable, Flexible, and Stretchable Strain/Pressure Sensors

Hardness and rigidity of micro and smart electronics cause mechanical deformation because of brittleness and inflexibility of their material components. This has restricted the application of non-flexible electronics that were replaced with electronic devices containing polymer and plastic substrates as flexible electronics in the past decades. In electronic devices, flexibility implies lightweight, elasticity, non-brittle, ductility, and easy fabrication [47]. Flexible electronics aroused great interest among researchers, industries, organizations, and military sectors to design and manufacture hybrid composite structures with lightweight flexible electronic devices [48]. On the other hand, the rapid growth of technologies, especially Internet in our daily life, has resulted in the advent of wearable technology in recent years [49]. Wearable and flexible sensors and biosensors have been widely explored in research and industry [50–52]. Sensors convert one form of energy into another form for processing [53]. Sensors detect environmental changes using conductive materials and send data to the computer for processing [54]. Conventional detection methods have been replaced by wearable and stretchable sensing electronics with nanomaterials [55]. Structures and designs of wearable sensors as well as nanomaterials determine sensor applications and their performance. Nanomaterials and nanocomposites depending on their properties, shape, size, structure, porosity, and fabrication method impart different sensor properties to wearable substrates. Strain/pressure sensors measure deformations, pressure stimuli, and mechanical properties [56, 57]. Metallic and semiconducting films were conventionally used as strain sensors and exhibited brittleness and rigidity with low stretchability [58]. These bulk materials could not attach to wearable substrates.

Nanomaterials including metal and metal-oxide nanoparticles and nanowires, conducting polymers (CPs), graphene, and CNTs have been explored due to their larger surface area compared to bulk materials. Superior mechanical and electrical properties and use of lower active materials are some of the advantages of the

integration of nanostructures and hybrid nanomaterials with flexible supports. The supports can also maintain their stretchability, wearability, and flexibility [59–64]. Flexible wearable piezoresistive strain/pressure sensors are composed of conductive sensing elements in addition to elastic polymers or other flexible/stretchable substrates. The contact resistance of the conductive part changes with strain/pressure stimuli, detected by electrical measuring systems. Different conductive nanomaterials have been explored as conductive fillers of elastomer composites including metal nanoparticles, nanowires and thin films, CPs, and carbon nanomaterials due to their desirable electrical and mechanical properties [65–71]. PDMS, polyurethane (PU), silicon rubber, Ecoflex, and Lycra are also explored as elastic substrates or components for conductive nanomaterials [72, 73]. On the other hand, incorporation of metallic nanomaterials, CNTs, and graphene with flexible wearable substrates propose desirable gauge factor ranges, introducing different structures and resulting in strains inside the matrix during stretching [74, 75]. Disconnection between nanomaterials decrease the contact area between them, leading to higher electrical resistance.

Graphene is another impressive carbon-based nanomaterial for resistance-based strain sensors due to its high electrical conductivity and mechanical properties [76–78]. Applied strain can change and partially destroy the hexagonal structure of the edges of graphene films [79]. Graphene can be combined with elastic polymers, which are generally grown through chemical vapor deposition or derived from exfoliated graphite. The shape and structure of graphene determine the flexibility and stretchability of sensors. To increase the stretchability and sensitivity of graphene in strain/pressure sensors, certain structures and engineering strategies were designed [80, 81]. In another work, macroscopic, flat, uniform graphene ribbons were fabricated through a modified wet spinning method [82]. Carbon black (CB) and graphite were also integrated with textiles to show piezoresistivity [83–85]. Natural biomaterials including natural textiles such as silk and cotton can be converted into conductive carbon materials via hydrothermal and pyrolysis processes [86, 87]. Carbonized silk nanofiber membranes integrated with PDMS film were also reported as skin-like pressure sensors.

1.6 Conductive Nanomaterials for Printed and Flexible Electronics Application

Printing is a process whereby formulated inks are deposited on the top of the substrate material for data transfer just as graphics, images, text, and the forms reappear as an image-carrying medium called substrate. Accordingly, printed and flexible electronics follow a set of printing methodologies to build electronic devices on flexible substrates. In the bubble-jet printing model, there is a print head with a heating unit. This heating unit is responsible for the preparation of ink bubbles and also forces the ink drop out of the nozzle. In the piezoelectric printing model, the nozzle

has a ceramic piezoelectric tile that gets expanded into the ink reservoir inside the print head and forces the ink drop out of the nozzle. The technology that enables to print inks on a variety of substrates with flexible or rigid structure materials is called screen printing. The entire process is relatively simple, flexible, and economical. The ink passes through a stencil made of a porous material during printing and gets deposited onto the substrate material. A delicate mesh made of nylon is used as a whole screen and stretched over a wooden or aluminum frame with a fitness of 150–400 fiber in.⁻¹

A high-resolution printing technique that uses an electric field to produce an essential fluid flow for ink delivery onto a substrate is called an electrohydrodynamic printing. It produces two to five smaller droplets compared to the nozzle size. The positive impact on the shape regulation of nanoparticles in the droplets developed for printing is due to the electric field existing between the substrate and the nozzle. It is also preferred in 3D structure additive manufacturing. Magdassi et al. prepared concentrated citrate-reduced silver nanocolloids as ink for inkjet printing on surfaces such as glass and paper. The material was prepared by redispersion following lyophilization of newly prepared colloids. Carboxymethyl cellulose, a polymer stabilizer, was used as a binder and stabilizer. The material was characterized by its high electrical conductivity, reactivity, and optical properties [88]. Shankar et al. [89] synthesized Ag NPs by a reduction method using tin acetate as a reducing agent with a passivating shell of dodecylamine (DDA). This process was carried out in the toluene medium. The aim was to formulate nonaqueous conductive ink to print circuits for antennas and radio frequency identification (RFID) tags on surfaces such as Epson photo paper through aerosol jet printing. Lai et al. presented a review article on synthesis of silver NP and inks. Also, control of chemical concentration is significant to maintain the size and morphology of nanoparticles. The most suggested method to synthesize nanoparticles is chemical reduction, which requires a reducing agent to decrease the ion solution to the physical nanoparticle form. The authors concluded that most printing metal inks in the market are silver based owing to its remarkable electrical properties [90]. Zhou et al. performed a study on DDA-protected Ag NPs by modifying a small amount of dodecanethiol as the co-protective agent to formulate conductive silver ink for inkjet printing. The study revealed that electrical properties of silver can be enhanced quickly by heat treating at 90 °C, thus proving its suitability to print electronic circuits [91]. A similar team also synthesized water-dispersible Ag NPs using polyacrylic acid as a capping agent to inkjet print on paper substrates. Following the treatment of the Ag NPs with the poly(diallyldimethylammonium chloride) [PDAC] solution at room temperature, the nanoparticles coalesced together after the detachment of polyacrylic acid (PAA). The films showed changes in the morphology and electrical properties. Which is significant identification for printed electronic devices [92]. Zhang et al. synthesized Ag NPs for paper-based electronics through inkjet printing. Conductivity of the printed silver tracks was enhanced by a multiple-step room temperature (RT) post-processing method. Electrical properties, mechanical flexibility, and strain sensitivity of the material are proven effective to serve the 3D

electronics in future [93]. Kastner et al. and Shahariar developed Ag NPs for inkjet printing of conductive lines on textiles. The reactive silver ink contained silver in the complexed form, and no dispersions were found. The inkjet printer electronic circuit paths on the fabric were found to withstand washing tests, moisture, and heat [94, 95]. Liang et al. developed water-based Ag nanowires to fabricate wearable thin-film transistors and stretchable conductors through screen printing on polyethylene terephthalate (PET) sheets. A new water-based Ag nanowire ink was prepared using a mixture of (hydroxypropyl) methylcellulose, fluorosurfactant, distilled water (DI) as a solvent, and an antifoaming agent as an organic binder. Investigations were carried out to connect the rheological properties of the ink with its composition [96].

Patil et al. performed a study on inkjet printing of silver nanowires synthesized by the polyol method. The study focused on conductivity improvement printed pattern stability on flexible surfaces and techniques followed. The process was optimized by varying the concentration of isopropyl alcohol and the Ag nanowire mixture suspended in ethylene glycol, which was synthesized. The printed patterns obtained using the developed ink demonstrated high conductivity and strength [97]. Zhao et al. performed their study on selective dopamine determination in human serum using inkjet-printed multi-walled CNT chips. The CNT ink was examined with homogeneous double layers to increase the efficiency of dopamine detection. This method successfully resulted in the demonstration of the samples. The low-cost approach for the ink formulation paves a way for futuristic applications in printed sensors [98].

1.7 Metal-oxide Semiconductors for Noninvasive Diagnosis of Breast Cancer

In recent years, research in biomedical engineering has enabled clinicians to achieve early medical diagnosis with help of advanced technologies. Although these conventional diagnostic techniques help in diagnosis of cancer, they have certain drawbacks: high cost, painful to patients, and may not always give accurate results, especially in the early stage of the disease.

Recent research has revealed that the overall cost of cancer diagnosis and treatment is indeed decreasing for patients treated in the early stages; hence, the prognostic management of cancer remains a topic of key interest in the scientific community, from fabrication of devices to even machine learning [99]. Over the past few decades, advances in molecular biology and biotechnologies have achieved significant milestones to identify cellular changes at the molecular level, which may be a promising approach for early disease detection. Moreover, present technologies can quantify cellular changes at an early phase of disease development. This may help in identifying particular pathophysiological processes and underlying conditions to build more accurate devices [100–103]. The early diagnosis of cancer is crucial for patient survival; therefore, sensitive and specific methods are required for early cancer diagnosis. These methods aid clinicians in successfully diagnosing and treating patients, thus, increasing their chances of survival [104].

The conventional diagnostic methods for cancer detection involve histological characterization such as staining based on cell morphology and analyzing the slides using microscopic techniques. These are invasive techniques that require human sample collection by taking a biopsy sample followed by sample processing techniques that involve cell fixation and further analysis using a microscope to detect cancer cells [105–107]. Breast cancer is a result of interplay of various factors that lead to genetic mutations. Hereditary and environmental factors play a major role in causing these mutations, which inactivate the DNA repair genes in the breast cells [108]. The main biomarkers associated with targeted therapy of breast cancer are estrogen receptor and progesterone receptor. Growth factors such as epidermal growth factors, mainly ErbB-1 (EGFR) and ErbB-2 (ERBB2, HER-2/neu, HER2), are also regulated in breast cancer tumorigenesis [109, 110]. EGFR overexpression is correlated to overexpression of HER-2, levels of which are used for targeted therapy by using the drug trastuzumab [111]. However, breast cancer type is classified based on gene expression profiling, which requires a large amount of tumor tissue for analysis and is highly expensive and also a difficult procedure. Moreover, gene expression in an individual is subject to changes due to post-transcriptional modifications and may lead to upregulation or downregulation of certain genes [112]. The diagnosis of breast-cell-related diseases requires an extensive biopsy and sampling for confirmation. These invasive diagnostic techniques cause distress until the diagnosis is confirmed through biopsy, and sometimes even after treatment there is a high chance of relapse; therefore, the patient has to get a biopsy done every few months. Some benign tumors, such as premalignant hyperplasia, when left untreated can also lead to invasive breast cancer, which is one of the most common malignancies diagnosed in women.

Breath analysis is widely being explored as a noninvasive diagnostic test for detection of breast cancer [113]. Increased oxidative stress is a key factor regulated in breast cancer, which causes damage to the cellular DNA, lipids, and proteins. Also, cytochrome P450 enzyme, expressed in breast cancer, causes lipid peroxidation of polyunsaturated fatty acids and leads to overexpression of volatile organic compounds such as alkanes and alkane derivatives in breath [114, 115]. Apart from lipid peroxidation, free radical-induced oxidation of certain amino acids, proteins, and peptides has also produced hydrocarbons and monoaldehydes [116, 117]. Recently, implementation of optical fibers in sensor application is rapidly increasing due to their ease of fabrication, low cost, and immunity to electromagnetic interferences. Especially in volatile organic compound (VOC) gas sensor applications, optical fibers are widely used because of their fast response and sensitivity toward gas molecules. Different types of optical fibers have been reported in sensor applications such as plastic optical fibers, silica optical fibers, fiber Bragg grating optical fibers [118], photonic crystal optical fibers [119]. In addition to the geometry and types of optical fibers, various sensing mechanisms have been reported, including absorption-based sensors, evanescent mode [120], plasmonic mode optical sensors such as surface plasmon resonance [121], localized surface plasmon resonance [122], and lossy mode resonance [123].

1.8 Down-conversion Photoluminescence Properties of $\text{ZrO}_2\text{:Ln}^{3+}$ (Ln = Eu, Sm, Er, Tb, Ho, Tm, Pr, Gd, Dy) Films Formed by Plasma Electrolytic Oxidation

In recent years, plasma electrolytic oxidation (PEO) is a very convenient process for formation of Ln ion-doped metal-oxide semiconductor films such as ZrO_2 [124–132], Al_2O_3 [133–139], TiO_2 [140, 141], Zn [142], Nb_2O_5 [143], HfO_2 [144], Y_2O_3 [145], and Gd_2O_3 [146] by PEO of pure Zr, Al, Ti, Zn, Nb, Hf, Y, and Gd metals, respectively, in an electrolyte containing suitable lanthanide oxide (Ln_2O_3) particles. The formation mechanism of the PEO film is complex, as it involves electro-, thermo-, and plasma-chemical reactions in the electrolyte and metal–electrolyte interface, due to locally high temperature (10^3 – 10^4 K) and pressure ($\sim 10^2$ MPa) at the micro-discharge sites [147].

During the PEO process of zirconium in an alkaline electrolyte containing Ln_2O_3 particles, ZrO_2 films are formed by PEO as a result of the oxidation of zirconium substrate via a sequence of localized micro-discharging events. Zr^{4+} ions formed at the zirconium/oxide interface ($\text{Zr} \rightarrow \text{Zr}^{4+} + 4\text{e}^-$) enter the micro-discharge channels and react with $\text{O}^{2-}/\text{OH}^-$ ions from the electrolyte. Metal oxides are eminent host matrices for Ln^{3+} due to their outstanding physicochemical characteristics such as good thermal, chemical, and physical stability, wide bandgap, broad optical transparency, low phonon frequency. Among them, zirconium oxide (zirconia, ZrO_2) is an excellent host material because its low phonon energy of about 470 cm^{-1} can increase the probability of radiative transition occurrence and reduce non-radiative multiphonon relaxation of excited Ln^{3+} throughout the vibrational bands of the ZrO_2 host lattice, thus foreseeing a high yield of intra-4f transitions from the Ln^{3+} [148–150].

The Ln_2O_3 particles take part in PEO process through the electrophoretic and micro-discharging mechanisms [151]. Ln_2O_3 particles have a negative zeta potential in alkaline media; thus, its particles are negatively charged. Due to the electrophoretic effect, the Ln_2O_3 particles move toward the zirconium anode. Melting point of Ln_2O_3 particles ($\sim 2400^\circ\text{C}$) is much lower than locally high temperatures ($\sim 7000^\circ\text{C}$) which induced at the micro-discharging sites [152], allowing the molten Ln_2O_3 particles to react with ZrO_2 and form Ln^{3+} ion-doped ZrO_2 films.

1.9 Multiferroics for Spintronic Applications

Multiferroics are a special class of materials that simultaneously possess two or more primary ferroic order parameters in the same phase such as ferroelectricity, ferromagnetism, ferroelasticity, and ferrotoroidicity [153]. In the last few decades, a significant number of research works have been done on multiferroics. A material that simultaneously possesses multiferroic orders is not sufficient. There should be a strong coupling interaction between different orders to achieve potential applications. Magnetoelectric coupling refers to the linear magnetoelectric effect [154]. One

can manipulate electric polarization by a magnetic field and magnetization by an electric field [155], which makes them potential candidates in device fabrication for different applications – memory devices [156–158], spintronics, sensors [159, 160], etc. Spintronics, spin-based electronics, is a multidisciplinary and fast-developing research area. Spintronic research is closely related to physics, chemistry, materials science, electrical engineering, and computer science. In spintronic devices, electron spin is employed in addition to electron charge, whereas in conventional electronic devices electron charge is used to transfer and store data. The properties such as intrinsic non-volatility, smaller size, high speed, ultralow-power consumption make spin-based devices a potential candidate in the microelectronics industry.

Spintronics is the study of the role played by electron spin in solid-state physics and in the fabrication of possible devices that explicitly exploit spin properties in addition to charge degrees of freedom. Numerous experiments have been carried out using polarized neutron reflectometry (PNR), which clearly illustrates Fe/Cr superlattices that lead to the giant magnetoresistance (GMR) effect [161], but PNR has some limitations too [162]. In 1994, Daughton et al. reported the design of a magnetic sensor to detect very weak magnetic fields, which was considered the first GMR-based device [163].

The major achievement of GMR in read heads for magnetic hard-disk drives was made by IBM in 1997 [164]. The innovation of GMR-based devices has dramatically increased the storage capacity of hard disks from 1 to 20 gigabytes. Nowadays all conventional electronic devices are based on the electron flow, which is considered a binary system (on/off state). If electrons flow in a circuit, it is the on state, and if there is no electron flow, it is the off state. Researchers have been focusing on the fabrication of new spintronic devices that offer an increase in the power of device up to 10 000 times compared to conventional electronic devices. The magnetic materials required for the fabrication of spintronic devices should have a stable structure, good electrical properties, and the ability to be applied in thin film form. A novel material ($\text{Fe}_{0.66}\text{Dy}_{0.24}\text{Tb}_{0.1}\text{O}_{7-x}$) (FDTO) with a semiconducting behavior, suitable electrical conductivity, and a high magnetic moment was reported, all of which can be controlled by altering the oxygen concentration during deposition. Recently, double-perovskite $\text{Pr}_2\text{FeCrO}_6$ (PFCO) was reported. Studies revealed that PFCO could be a promising multiferroic material for spintronic applications. The field dependence of magnetoresistance (MR) for PFCO showed MR% of 44% with a clear distinction of resistance states in parallel and antiparallel states between 0 and ± 5 KOe observed [165].

A considerable change in MR with field revealed that PFCO was a potential candidate for the magnetic tunnel junction (MTJ) in spintronics using multiferroic materials. Spin valves, fabricated in 1991 [166], led to a drastic change in magnetic sensors [167, 168], read heads of magnetic sensors [169, 170], and magnetic random-access memory (MRAM). The spin valve structure in either current-in-plane (CIP) or current-perpendicular-to-plane (CPP) form can be used to analyze the GMR effect.

1.10 Quartz Tuning Fork Based Nanosensors

Though the commercially available quartz tuning fork (QTF) was first designed as a timing crystal for wrist watches in the early 1970s, it held, and still holds, much promise as a scientific tool. However, it would be more than a decade before the QTF's potential as a sensor was explored.

QTFs are single-crystal quartz mechanical oscillators, which are relatively inexpensive and have been used in wrist watches and other high precision timing circuits for decades. Piezoelectricity of quartz, which enables the conversion of electrical signals into mechanical vibrations and vice versa, allows the QTFs to be excited electrically and their resonant frequencies to be read out electrically. These QTFs can be considered mechanical transducers since changes in the mechanical properties of a system can be monitored by measuring the changes in the resonant frequency of the system, which is measured in the form of an electrical signal. Owing to their high precision resonance frequency, high quality (Q) factor, high mechanical stability, near-zero thermal deflection, and low power consumption, they are excellent candidates for use in timing circuits, probe microscopy techniques, gyroscopes, micro balances, and sensors [171, 172]. The QTF was modified by surface coating the prongs with a thin layer of polystyrene (PS) by dipping the fork in a solution of PS and aniline and allowing aniline to evaporate off the fork. For human IgG detection, the PS-coated QTFs were again dipped in anti-human IgG and allowed to dry. For self-excited QTFs, the prongs were coated with varying concentrations of PS and their responses were measured.

In 2003, S. Boussaad and N. J. Tao published a landmark study “Polymer wire chemical sensor using a microfabricated tuning fork” in *Nano Letters*. This was a milestone in the development of QTFs as a nanosensor since the study explored the possibility of using more than simple mass loading to detect analytes. They successfully formed a bridge by attaching a nitrocellulose/toluene sulfonamide formaldehyde resin wire between the two tines of a QTF.

The QTFs were attached with polyvinylcinnamate (PVCN) wires, and the wires were crosslinked by exposing them to UV radiation for one to three minutes, thereby the wires could shrink and their stiffness increased. A comparison of the frequency responses of QTFs before and after crosslinking the wires showed that crosslinking increased the sensitivity and Q factor of the sensor by amplifying the change in the modulus of the wire due to the presence or absence of ethanol vapors [173].

In another study, published in 2016, the electrospun wires of polymethyl methacrylate (PMMA) onto the QTF and formed polymer bridges. Then, they exposed the sensors to water vapor in order to use them as humidity sensors. The resonance frequency of the sensors reduced with an increase in the humidity level due to adsorption of the water molecules onto the PMMA wires, leading to a decrease in the modulus of the polymer bridges. The amount of decrease, for a given humidity level, was found to be a function of the number of wires present [174].

References

- 1 Rodriguez-Perez, L., Herranz, M.Á., and Martin, N. (2013). The chemistry of pristine graphene. *Chem. Commun.* 49: 3721–3735.
- 2 Dreyer, D.R., Park, S., Bielawski, C.W., and Ruoff, R.S. (2010). The chemistry of graphene oxide. *Chem. Soc. Rev.* 39: 228–240.
- 3 Eigler, S. and Hirsch, A. (2014). Chemistry with graphene and graphene oxide – challenges for synthetic chemists. *Angew. Chem. Int. Ed.* 53: 7720–7738.
- 4 Bhorla, R.S. (2015). Graphene: potential material for nanoelectronics applications. *Indian J. Pure Appl. Phys.* 53: 501–513.
- 5 Li, Z., Ma, K., Cheng, Z. et al. (2017). Fabrication of electrochemical immunosensor for cardiac biomarker troponin I determination and its potential for acute myocardial infarction diagnosis. *Int. J. Electrochem. Sci.* 12: 2389–2399.
- 6 Bhatnagar, D., Kaur, I., and Kumar, A. (2017). Ultrasensitive cardiac troponin I antibody based nanohybrid sensor for rapid detection of human heart attack. *Int. J. Biol. Macromol.* 95: 505–510.
- 7 Qi, M., Zhang, Y., Cao, C. et al. (2016). Increased sensitivity of extracellular glucose monitoring based on AuNP decorated GO nanocomposites. *RSC Adv.* 6: 39180–39187.
- 8 Song, J., Xu, L., Zhou, C. et al. (2013). Synthesis of graphene oxide based CuO nanoparticles composite electrode for highly enhanced nonenzymatic glucose detection. *ACS Appl. Mater. Interfaces* 5: 12928–12934.
- 9 Wang, L., Lu, X., Wen, C. et al. (2015). One-step synthesis of Pt–NiO nanoplate array/reduced graphene oxide nanocomposites for nonenzymatic glucose sensing. *J. Mater. Chem. A* 3: 608–616.
- 10 Justice Babu, K., Sheet, S., Lee, Y.S., and Gnana Kumar, G. (2018). Three-dimensional dendrite Cu–Co/reduced graphene oxide architectures on a disposable pencil graphite electrode as an electrochemical sensor for nonenzymatic glucose detection. *ACS Sustain. Chem. Eng.* 6: 1909–1918.
- 11 Manthina, V. and Agrios, A.G. (2016). Single-pot ZnO nanostructure synthesis by chemical bath deposition and their applications. *Nano-Struct. Nano-Objects* 7: 1–11.
- 12 Ghayempour, S. and Montazer, M. (2017). Ultrasound irradiation based in-Situ synthesis of star-like tragacanth gum/zinc oxide nanoparticles on cotton fabric. *Ultrason. Sonochem.* 34: 458–465.
- 13 Gayen, R. and Paul, R. (2018). Phosphorous doping in vertically aligned ZnO nanorods grown by wet-chemical method. *Nano-Struct. Nano-Objects* 13: 163–169.
- 14 Fakhri, P., Amini, B., Bagherzadeh, R. et al. (2019). Flexible hybrid structure piezoelectric nanogenerator based on ZnO nanorod/Pvdf nanofibers with improved output. *RSC Adv.* 9: 10117–10123.

- 15 Masuda, Y., Kinoshita, N., and Koumoto, K. (2012). Liquid phase morphology control of ZnO nanowires, ellipse particles, hexagonal rods, and particle in aqueous solutions. *ISRN Nanotechnol.* 2012: 1–6.
- 16 Choi, M.Y., Choi, D., Jin, M.J. et al. (2009). Mechanically powered transparent flexible charge-generating nanodevices with piezoelectric ZnO nanorods. *Adv. Mater.* 21: 2185–2189.
- 17 Rodriguez-Paéz, J., Caballero, A., Villegas, M. et al. (2001). Controlled precipitation methods: formation mechanism of ZnO nanoparticles. *J. Eur. Ceram. Soc.* 21: 925–930.
- 18 Aladpoosh, R. and Montazer, M.J. (2015). The role of cellulosic chains of cotton in biosynthesis of ZnO nanorods producing multifunctional properties: mechanism, characterizations and features. *Carbohydr. Polym.* 126: 122–129.
- 19 Wang, Z.L. and Song, J. (2006). Piezoelectric nanogenerators based on zinc oxide nanowire arrays. *Science* 312: 242–246.
- 20 Wang, Z.L. (2009). Energy harvesting using piezoelectric nanowires – a correspondence on “energy harvesting using nanowires?” by Alexe et al. *Adv. Mater.* 21: 1311–1315.
- 21 Özgür, Ü., Alivov, Y.I., Liu, C. et al. (2005). A comprehensive review of ZnO materials and devices. *J. Appl. Phys.* 98: 11.
- 22 Moezzi, A., McDonagh, A.M., and Cortie, M.B. (2012). Zinc oxide particles: synthesis, properties and applications. *Chem. Eng. J.* 185: 1–22.
- 23 Xi, Y., Song, J., Xu, S. et al. (2009). Growth of ZnO nanotube arrays and nanotube based piezoelectric nanogenerators. *J. Mater. Chem.* 19: 9260–9264.
- 24 Savarimuthu, K., Rajamanickam, G., Shankararajan, R. et al. (2017). Experimental study on flexible ZnO based nanogenerator using Schottky contact for energy harvesting applications. *IEEE Trans. Nanotechnol.* 16: 469–476.
- 25 Takeda, E. (1993). Hot-carrier effects in scaled MOS devices, microelectronics reliability. *Microelectron. Reliab.* 33: 1687–1711.
- 26 Roblin, P., Samman, A., and Bibyk, S. (1988). Simulation of hot-electron trapping and aging of nMOSFETs. *IEEE Trans. Electron Devices* 35: 2229–2237.
- 27 Tam, S., Ko, P.K., and Hu, C. (1984). Lucky-electron model of channel hot-electron injection in MOSFET's. *IEEE Trans. Electron Devices* 31: 1116–1125.
- 28 Hänsch, W., Schwerin, A.V., and Hofmann, F. (1990). A new self-consistent modeling approach to investigating MOSFET degradation. *IEEE Electron Device Lett.* 11: 362–364.
- 29 Doyle, B., Bourcier, M., Marchetaux, J.C., and Boudou, A. (1990). Interface state creation and charge trapping in the medium-to-high gate voltage range ($V_d/2 \geq V_g \geq V_d$) during hot-carrier stressing of n-MOS transistors. *IEEE Trans. Electron Devices* 37: 744–754.
- 30 Chen, Z., Hess, K., Lee, J. et al. (2000). On the mechanism for interface trap generation in MOS transistors due to channel hot carrier stressing. *IEEE Electron Device Lett.* 21: 24–26.

- 31 A. Bravaix, C. Guerin, V. Huard et al. (2009). Hot-carrier acceleration factors for low power management in DC-AC stressed 40 nm NMOS node at high temperature. Reliability Physics Symposium, 2009 IEEE International, Montreal, QC, Canada, 26–30 April 2009.
- 32 Djeflal, F., Ghoggali, Z., Dibi, Z., and Lakhdar, N. (2009). Analytical analysis of nanoscale multiple gate MOSFETs including effects of hot-carrier induced interface charges. *Microelectron. Reliab.* 49: 377–381.
- 33 Rauch, S., La Rosa, G., and Guarin, F. (2001). Role of e-e scattering in the enhancement of channel hot carrier degradation of deep-submicron nMOSFETs at high V_{gs} conditions. *IEEE Trans. Device Mater. Reliab.* 1: 113–119.
- 34 Lee, C.W., Afzalian, A., Dehdashti Akhavan, N. et al. (2009). Junctionless multi-gate field-effect transistor. *Appl. Phys. Lett.* 94 (1–2): 053511.
- 35 Colinge, J.P., Lee, C.W., Afzalian, A. et al. (2010). Nanowire transistors without junctions. *Nat. Nanotechnol.* 5: 225–229.
- 36 Rios, R., Cappellani, A., Armstrong, M. et al. (Sep. 2011). Comparison of junctionless and conventional trigate transistors with L_g down to 26 nm. *IEEE Electron Device Lett.* 32: 1170–1172.
- 37 Jeon, D.Y. (Dec. 2013). Revisited parameter extraction methodology for electrical characterization of junctionless transistors. *Solid State Electron.* 90: 86–93.
- 38 Colinge, J.P., Lee, C.W., Dehdashti Akhavan, N. et al. (2012). Junctionless transistors: physics and properties. In: *Semiconductor-On-Insulator Materials for Nanoelectronics Applications* (ed. A. Nazarov, J.-P. Colinge, F. Balestra, et al.), 187–200. Berlin Heidelberg: Springer.
- 39 Goto, K.I., Yu, T.H., Wu, J. et al. (2012). Mobility and screening effect in heavily doped accumulation-mode metal-oxide-semiconductor field-effect transistors. *Appl. Phys. Lett.* 101: 073503.
- 40 Rudenko, T., Nazarov, A., Yu, R. et al. (Sep. 2013). Electron mobility in heavily doped junctionless nanowire SOI MOSFETs. *Microelectron. Eng.* 109: 326–329.
- 41 Tavakolian, F., Farrokhabadi, A., SoltanRezaee, M., and Rahmanian, S. (2019). Dynamic pull-in of thermal cantilever nanoswitches subjected to dispersion and axial forces using nonlocal elasticity theory. *Microsyst. Technol.* 25 (1): 19–30.
- 42 Esfahani, S., Khadem, S.E., and Mamaghani, A.E. (2019). Size-dependent nonlinear vibration of an electrostatic nanobeam actuator considering surface effects and inter-molecular interactions. *Int. J. Mech. Mater. Des.* 15 (3): 489–505.
- 43 SoltanRezaee, M., Bodaghi, M., Farrokhabadi, A., and Hedayati, R. (2019). Non-linear stability analysis of piecewise actuated piezoelectric microstructures. *Int. J. Mech. Sci.* 160: 200–208.
- 44 Lamoreaux, S.K. (2005). The Casimir force: background, experiments, and applications. *Rep. Prog. Phys.* 68 (1): 201.
- 45 Chan, H., Bao, Y., Zou, J. et al. (2008). Measurement of the Casimir force between a gold sphere and a silicon surface with nanoscale trench arrays. *Phys. Rev. Lett.* 101 (3): 030401.
- 46 Rahi, S.J., Emig, T., Jaffe, R.L. et al. (2008). Casimir forces between cylinders and plates. *Phys. Rev. A* 78 (1): 012104.

- 47 Wong, W.S. and Salleo, A. (2009). *Flexible Electronics: Materials and Applications*, vol. 11. Springer Science & Business Media.
- 48 Zuber, K., Fabretto, M., Hall, C., and Murphy, P. (2008). Improved PEDOT conductivity via suppression of crystallite formation in Fe (III) tosylate during vapor phase polymerization. *Macromol. Rapid Commun.* 29 (18): 1503–1508.
- 49 McCann, J. and Bryson, D. (2009). *Smart Clothes and Wearable Technology*. Elsevier.
- 50 Patel, S., Park, H., Bonato, P. et al. (2012). A review of wearable sensors and systems with application in rehabilitation. *J. Neuroeng. Rehabil* 9 (1): 21.
- 51 Sedighi, A., Montazer, M., and Mazinani, S. (2019). Synthesis of wearable and flexible NiP0. 1-SnOx/PANI/CuO/cotton towards a non-enzymatic glucose sensor. *Biosens. Bioelectron.* 135: 192–199.
- 52 Sedighi, A., Montazer, M., and Samadi, N. (2014). Synthesis of nano Cu₂O on cotton: morphological, physical, biological and optical sensing characterizations. *Carbohydr. Polym.* 110: 489–498.
- 53 Jayathilaka, W.A.D.M., Qi, K., Qin, Y. et al. (2019). Significance of nanomaterials in wearables: a review on wearable actuators and sensors. *Adv. Mater.* 31 (7): 1805921.
- 54 Wu, W. (2017). Inorganic nanomaterials for printed electronics: a review. *Nanoscale* 9 (22): 7342–7372.
- 55 Wang, X., Liu, Z., and Zhang, T. (2017). Flexible sensing electronics for wearable/attachable health monitoring. *Small* 13 (25): 1602790.
- 56 Lou, Z., Li, L., Wang, L., and Shen, G. (2017). Recent progress of self-powered sensing systems for wearable electronics. *Small* 13 (45): 1701791.
- 57 Wang, C., Xia, K., Wang, H. et al. (2019). Advanced carbon for flexible and wearable electronics. *Adv. Mater.* 31 (9): 1801072.
- 58 Wagner, S. and Bauer, S. (2012). Materials for stretchable electronics. *MRS Bull.* 37 (3): 207–213.
- 59 Yao, S., Swetha, P., and Zhu, Y. (2018). Nanomaterial-enabled wearable sensors for healthcare. *Adv. Healthc. Mater* 7 (1): 1700889.
- 60 Wang, Z., Zhou, H., Chen, W. et al. (2018). Dually synergetic network hydrogels with integrated mechanical stretchability, thermal responsiveness, and electrical conductivity for strain sensors and temperature alertors. *ACS Appl. Mater. Interfaces* 10 (16): 14045–14054.
- 61 Wang, Y., Hao, J., Huang, Z. et al. (2018). Flexible electrically resistive-type strain sensors based on reduced graphene oxide-decorated electrospun polymer fibrous mats for human motion monitoring. *Carbon* 126: 360–371.
- 62 Heo, Y., Hwang, Y., Jung, H.S. et al. (2017). Secondary sensitivity control of silver-nanowire-based resistive-type strain sensors by geometric modulation of the elastomer substrate. *Small* 13 (23): 1700070.
- 63 Zhong, J., Zhong, Q., Hu, Q. et al. (2015). Stretchable self-powered fiber-based strain sensor. *Adv. Funct. Mater.* 25 (12): 1798–1803.
- 64 Lee, J., Shin, S., Lee, S. et al. (2018). Highly sensitive multifilament fiber strain sensors with ultrabroad sensing range for textile electronics. *ACS Nano* 12 (5): 4259–4268.

- 65 Hou, C., Wang, H., Zhang, Q. et al. (2014). Highly conductive, flexible, and compressible all-graphene passive electronic skin for sensing human touch. *Adv. Mater.* 26 (29): 5018–5024.
- 66 Yao, H.B., Ge, J., Wang, C.F. et al. (2013). A flexible and highly pressure-sensitive graphene–polyurethane sponge based on fractured microstructure design. *Adv. Mater.* 25 (46): 6692–6698.
- 67 Zhu, B., Niu, Z., Wang, H. et al. (2014). Microstructured graphene arrays for highly sensitive flexible tactile sensors. *Small* 10 (18): 3625–3631.
- 68 Jung, S., Kim, J.H., Kim, J. et al. (2014). Reverse-micelle-induced porous pressure-sensitive rubber for wearable human–machine interfaces. *Adv. Mater.* 26 (28): 4825–4830.
- 69 Park, J., Lee, Y., Hong, J. et al. (2014). Tactile-direction-sensitive and stretchable electronic skins based on human-skin-inspired interlocked microstructures. *ACS Nano* 8 (12): 12020–12029.
- 70 Choong, C.L., Shim, M.B., Lee, B.S. et al. (2014). Highly stretchable resistive pressure sensors using a conductive elastomeric composite on a micropylamrid array. *Adv. Mater.* 26 (21): 3451–3458.
- 71 Gong, S., Schwalb, W., Wang, Y. et al. (2014). A wearable and highly sensitive pressure sensor with ultrathin gold nanowires. *Nat. Commun.* 5 (1): 1–8.
- 72 Bae, S.-H., Lee, Y., Sharma, B.K. et al. (2013). Graphene-based transparent strain sensor. *Carbon* 51: 236–242.
- 73 Zhang, F., Zang, Y., Huang, D. et al. (2015). Flexible and self-powered temperature–pressure dual-parameter sensors using microstructure-frame-supported organic thermoelectric materials. *Nat. Commun.* 6 (1): 1–10.
- 74 Gong, S., Lai, D.T., Su, B. et al. (2015). Highly stretchy black gold E-skin nanopatches as highly sensitive wearable biomedical sensors. *Adv. Electron.* 1 (4): 1400063.
- 75 Gong, S., Lai, D.T., Wang, Y. et al. (2015). Tattoolike polyaniline microparticle-doped gold nanowire patches as highly durable wearable sensors. *ACS Appl. Mater. Interfaces* 7 (35): 19700–19708.
- 76 Bae, G.Y., Pak, S.W., Kim, D. et al. (2016). Linearly and highly pressure-sensitive electronic skin based on a bioinspired hierarchical structural array. *Adv. Mater.* 28 (26): 5300–5306.
- 77 Boland, C.S., Khan, U., Backes, C. et al. (2014). Sensitive, high-strain, high-rate bodily motion sensors based on graphene–rubber composites. *ACS Nano* 8 (9): 8819–8830.
- 78 Yang, T., Jiang, X., Zhong, Y. et al. (2017). A wearable and highly sensitive graphene strain sensor for precise home-based pulse wave monitoring. *ACS Sens.* 2 (7): 967–974.
- 79 Li, Y., Jiang, X., Liu, Z., and Liu, Z. (2010). Strain effects in graphene and graphene nanoribbons: the underlying mechanism. *Nano Res.* 3 (8): 545–556.

- 80 Tian, H., Shu, Y., Cui, Y.-L. et al. (2014). Scalable fabrication of high-performance and flexible graphene strain sensors. *Nanoscale* 6 (2): 699–705.
- 81 Hempel, M., Nezich, D., Kong, J., and Hofmann, M. (2012). A novel class of strain gauges based on layered percolative films of 2D materials. *Nano Lett.* 12 (11): 5714–5718.
- 82 Sun, J., Li, Y., Peng, Q. et al. (2013). Macroscopic, flexible, high-performance graphene ribbons. *ACS Nano* 7 (11): 10225–10232.
- 83 Wang, F., Zhu, B., Shu, L., and Tao, X. (2013). Flexible pressure sensors for smart protective clothing against impact loading. *Smart Mater. Struct.* 23 (1): 015001.
- 84 Shyr, T.-W., Shie, J.-W., and Jhuang, Y.-E. (2011). The effect of tensile hysteresis and contact resistance on the performance of strain-resistant elastic-conductive webbing. *Sensors* 11 (2): 1693–1705.
- 85 Cochrane, C., Koncar, V., Lewandowski, M., and Dufour, C. (2007). Design and development of a flexible strain sensor for textile structures based on a conductive polymer composite. *Sensors* 7 (4): 473–492.
- 86 Li, Y., Samad, Y.A., and Liao, K. (2015). From cotton to wearable pressure sensor. *J. Mater. Chem. A* 3 (5): 2181–2187.
- 87 Wang, C., Li, X., Gao, E. et al. (2016). Carbonized silk fabric for ultrastretchable, highly sensitive, and wearable strain sensors. *Adv. Mater.* 28 (31): 6640–6648.
- 88 S. Magdassi, A. Bassa, Y. Vinetsky, and A. Kamyshny, “Silver nanoparticles as pigments for water-based ink-jet inks,” *Chem. Mater.*, vol. 15, no. 11, pp. 2208–2217, 2003, doi:<https://doi.org/10.1021/cm021804b>.
- 89 Shankar, R., Groven, L., Amert, A. et al. (2011). Non-aqueous synthesis of silver nanoparticles using tin acetate as a reducing agent for the conductive ink formulation in printed electronics. *J. Mater. Chem.* 21 (29): 10871–10877.
- 90 Lai, C.Y., Cheong, C.F., Mandeep, J.S. et al. (2014). Synthesis and characterization of silver nanoparticles and silver inks: review on the past and recent technology roadmaps. *J. Mater. Eng. Perform.* 23 (10): 3541–3550.
- 91 Zhou, X., Li, W., Wu, M. et al. (2014). Enhanced dispersibility and dispersion stability of dodecylamine-protected silver nanoparticles by dodecanethiol for ink-jet conductive inks. *Appl. Surf. Sci.* 292: 537–543.
- 92 Huang, Q., Shen, W., Xu, Q. et al. (2014). Room-temperature sintering of conductive Ag films on paper. *Mater. Lett.* 123: 124–127.
- 93 Zhang, X.Y., Xu, J.J., Wu, J.Y. et al. (2017). Seeds triggered massive synthesis and multi-step room temperature post-processing of silver nanoink – application for paper electronics. *RSC Adv.* 7 (1): 8–19.
- 94 Kastner, J., Faury, T., Außerhuber, H.M. et al. (2017). Silver-based reactive ink for inkjet-printing of conductive lines on textiles. *Microelectron. Eng.* 176: 84–88.
- 95 Shahariar, H., Kim, I., Soewardiman, H., and Jur, J.S. (2019). Inkjet printing of reactive silver ink on textiles. *ACS Appl. Mater. Interfaces* 11 (6): 6208–6216.

- 96 Liang, J., Tong, K., and Pei, Q. (2016). A water-based silver-nanowire screen-print ink for the fabrication of stretchable conductors and wearable thin-film transistors. *Adv. Mater.* 28 (28): 5986–5996.
- 97 P. Patil, S. Patil, P. Kate, and A. A. Kulkarni, “Inkjet printing of silver nanowires on flexible surfaces and methodologies to improve the conductivity and stability of the printed patterns,” *Nanoscale Adv.*, 2021, 31(1), 240–248.
- 98 Zhao, J., Yu, Y., Weng, B. et al. (2013). Sensitive and selective dopamine determination in human serum with inkjet printed Nafion/MWCNT chips. *Electrochem. Commun.* 37: 32–35.
- 99 Kourou, K., Exarchos, T.P., Exarchos, K.P. et al. (2015). Machine learning applications in cancer prognosis and prediction. *Comput. Struct. Biotechnol. J.* 13: 8–17.
- 100 Solassol, J., Marin, P., Demetree, E. et al. (2005). Proteomic detection of prostate-specific antigen using a serum fractionation procedure: potential implication for new low-abundance cancer biomarkers detection. *Anal. Biochem.* 338 (1): 26–31.
- 101 Singh, S., Gill, A.A.S., Nlooto, M., and Karpoomath, R. (2019). Prostate cancer biomarkers detection using nanoparticles based electrochemical biosensors. *Biosens. Bioelectron.* 137: 213–221.
- 102 Singh, P., Pandey, S.K., Singh, J. et al. (2016). Biomedical perspective of electrochemical nanobiosensor. *Nano-Micro Lett.* 8 (3): 193–203.
- 103 Wei, F., Patel, P., Liao, W. et al. (2009). Electrochemical sensor for multiplex biomarkers detection. *Clin. Cancer Res. An Off. J. Am. Assoc. Cancer Res.* 15 (13): 4446–4452.
- 104 Tripathi, P.P., Arami, H., Banga, I. et al. (2018). Cell penetrating peptides in preclinical and clinical cancer diagnosis and therapy. *Oncotarget* 9 (98): 37252–37267.
- 105 Alturkistani, H.A., Tashkandi, F.M., and Mohammedsaleh, Z.M. (2015). Histological stains: a literature review and case study. *Glob. J. Health Sci.* 8 (3): 72–79.
- 106 Zeng, C., Vangveravong, S., Xu, J. et al. (2007). Subcellular localization of sigma-2 receptors in breast cancer cells using two-photon and confocal microscopy. *Cancer Res.* 67 (14): 6708–6716.
- 107 Kaufmann, R., Müller, P., Hildenbrand, G. et al. (2011). Analysis of Her2/neu membrane protein clusters in different types of breast cancer cells using localization microscopy. *J. Microsc.* 242 (1): 46–54.
- 108 DeVita, V.T., Lawrence, T.S., and Rosenberg, S.A. (2008). *DeVita, Hellman, and Rosenberg’s Cancer : Principles & Practice of Oncology*. Philadelphia, PA: Wolters Kluwer/Lippincott Williams & Wilkins.
- 109 Cleator, S., Heller, W., and Coombes, R.C. (2007). Triple-negative breast cancer: therapeutic options. *Lancet. Oncol.* 8 (3): 235–244.
- 110 Phillips, M., Cataneo, R.N., Ditkoff, B.A. et al. (2003). Volatile markers of breast cancer in the breath. *Breast J.* 9 (3): 184–191.
- 111 Dean-Colomb, W. and Esteva, F.J. (2008). Her2-positive breast cancer: herceptin and beyond. *Eur. J. Cancer* 44 (18): 2806–2812.

- 112 Kodadek, T. (2001). Protein microarrays: prospects and problems. *Chem. Biol.* 8 (2): 105–115.
- 113 Shuster, G., Gallimidi, Z., Reiss, A.H. et al. (2011). Classification of breast cancer precursors through exhaled breath. *Breast Cancer Res. Treat.* 126 (3): 791–796.
- 114 Kasapović, J., Pejić, S., Stojilković, V. et al. (2010). Antioxidant status and lipid peroxidation in the blood of breast cancer patients of different ages after chemotherapy with 5-fluorouracil, doxorubicin and cyclophosphamide. *Clin. Biochem.* 43 (16–17): 1287–1293.
- 115 Mense, S.M., Remotti, F., Bhan, A. et al. (2008). Estrogen-induced breast cancer: alterations in breast morphology and oxidative stress as a function of estrogen exposure. *Toxicol. Appl. Pharmacol.* 232 (1): 78–85.
- 116 Kessler, W. and Remmer, H. (1990). Generation of volatile hydrocarbons from amino acids and proteins by an iron/ascorbate/GSH system. *Biochem. Pharmacol.* 39 (8): 1347–1351.
- 117 Clemens, M.R., Remmer, H., and Waller, H.D. (1984). Phenylhydrazine-induced lipid peroxidation of red blood cells in vitro and in vivo: monitoring by the production of volatile hydrocarbons. *Biochem. Pharmacol.* 33 (11): 1715–1718.
- 118 Jiang, T., Ren, L., Jia, Z. et al. (2017). Application of FBG based sensor in pipeline safety monitoring. *Appl. Sci.* 7 (6): 540.
- 119 Lopez-Torres, D., Elosua, C., and Arregui, F.J. (2020). Optical fiber sensors based on microstructured optical fibers to detect gases and volatile organic compounds – a review. *Sensors* 20 (9): 2555.
- 120 Narasimman, S., Balakrishnan, L., Meher, S.R. et al. (2017). Influence of surface functionalization on the gas sensing characteristics of ZnO nanorhombuses. *J. Alloys Compd.* 706: 186–197.
- 121 Lee, K.H., Zhang, Y.-Z., Jiang, Q. et al. (2020). Ultrasound-driven two-dimensional $\text{Ti}_3\text{C}_2\text{T}_x$ MXene hydrogel generator. *ACS Nano* 14 (3): 3199–3207.
- 122 Baldini, F., Brenci, M., Chiavaioli, F. et al. (2012). Optical fibre gratings as tools for chemical and biochemical sensing. *Anal. Bioanal. Chem.* 402 (1): 109–116.
- 123 Wang, Q. and Zhao, W.-M. (2018). A comprehensive review of lossy mode resonance-based fiber optic sensors. *Opt. Lasers Eng.* 100: 47–60.
- 124 Ćirić, A. and Stojadinović, S. (2019). Photoluminescence properties of Pr^{3+} doped ZrO_2 formed by plasma electrolytic oxidation. *J. Alloys Compd.* 803: 126–134.
- 125 Stojadinović, S. and Vasilić, R. (2016). Formation and photoluminescence of Eu^{3+} doped zirconia coatings formed by plasma electrolytic oxidation. *J. Lumin.* 176: 25–31.
- 126 Stojadinović, S., Tadić, N., and Vasilić, R. (2016). Photoluminescence of Sm^{3+} doped ZrO_2 coatings formed by plasma electrolytic oxidation of zirconium. *Mater. Lett.* 164: 329–332.
- 127 Stojadinović, S., Tadić, N., and Vasilić, R. (2019). Down- and up-conversion photoluminescence of $\text{ZrO}_2 : \text{Ho}^{3+}$ and $\text{ZrO}_2 : \text{Ho}^{3+}/\text{Yb}^{3+}$ coatings formed by plasma electrolytic oxidation. *J. Alloys Compd.* 785: 1222–1232.

- 128 Ćirić, A. and Stojadinović, S. (2019). Photoluminescence studies of $\text{ZrO}_2 : \text{Tm}^{3+}/\text{Yb}^{3+}$ coatings formed by plasma electrolytic oxidation. *J. Lumin.* 214: 116568.
- 129 Stojadinović, S., Tadić, N., and Vasilić, R. (2018). Down-conversion photoluminescence of $\text{ZrO}_2 : \text{Er}^{3+}$ coatings formed by plasma electrolytic oxidation. *Mater. Lett.* 219: 251–255.
- 130 Stojadinović, S., Tadić, N., and Vasilić, R. (2018). Structural and photoluminescent properties of $\text{ZrO}_2 : \text{Tb}^{3+}$ coatings formed by plasma electrolytic oxidation. *J. Lumin.* 197: 83–89.
- 131 Ćirić, A. and Stojadinović, S. (2020). Photoluminescence of $\text{ZrO}_2 : \text{Gd}^{3+}$ and $\text{ZrO}_2 : \text{Dy}^{3+}$ coatings formed by the plasma electrolytic oxidation. *J. Alloys Compd.* 832: 154907.
- 132 Stojadinović, S., Tadić, N., and Vasilić, R. (2019). Photoluminescence properties of $\text{Er}^{3+}/\text{Yb}^{3+}$ doped ZrO_2 coatings formed by plasma electrolytic oxidation. *J. Lumin.* 208: 296–301.
- 133 Stojadinović, S. and Vasilić, R. (2018). Eu^{2+} photoluminescence in Al_2O_3 coatings obtained by plasma electrolytic oxidation. *J. Lumin.* 199: 240–244.
- 134 Stojadinović, S., Tadić, N., and Vasilić, R. (2017). Photoluminescence of $\text{Sm}^{2+}/\text{Sm}^{3+}$ doped Al_2O_3 coatings formed by plasma electrolytic oxidation of aluminum. *J. Lumin.* 192: 110–116.
- 135 Stojadinović, S. and Vasilić, R. (2018). Photoluminescence of Ce^{3+} and $\text{Ce}^{3+}/\text{Tb}^{3+}$ ions in Al_2O_3 host formed by plasma electrolytic oxidation. *J. Lumin.* 203: 576–581.
- 136 Stojadinović, S. and Ćirić, A. (2019). $\text{Ce}^{3+}/\text{Eu}^{2+}$ doped Al_2O_3 coatings formed by plasma electrolytic oxidation of aluminum: photoluminescence enhancement by $\text{Ce}^{3+} \rightarrow \text{Eu}^{2+}$ energy transfer. *Coatings* 9: 819.
- 137 Stojadinović, S. and Ćirić, A. (2020). Dy^{3+} and $\text{Dy}^{3+}/\text{Ce}^{3+}$ doped Al_2O_3 coatings obtained by plasma electrolytic oxidation: Photoluminescence and energy transfer from Ce^{3+} to Dy^{3+} . *J. Lumin.* 226: 117403.
- 138 Stojadinović, S. and Ćirić, A. (2020). Highly-increased photoluminescence of Pr^{3+} due to the $\text{Eu}^{2+} \rightarrow \text{Pr}^{3+}$ energy transfer in Al_2O_3 coatings formed by plasma electrolytic oxidation. *J. Lumin.* 226: 117407.
- 139 Stojadinović, S. and Ćirić, A. (2021). Sensitizing effect of Ce^{3+} on Tm^{3+} blue emission in $\text{Al}_2\text{O}_3:\text{Tm}^{3+}/\text{Ce}^{3+}$ coatings formed by plasma electrolytic oxidation. *J. Mater. Sci. Mater. Electron.* 32: 7841–7847.
- 140 Stojadinović, S., Radić, N., Grbić, B. et al. (2016). Structural, photoluminescent and photocatalytic properties of $\text{TiO}_2:\text{Eu}^{3+}$ coatings formed by plasma electrolytic oxidation. *Appl. Surf. Sci.* 370: 218–228.
- 141 Stojadinović, S., Tadić, N., Radić, N. et al. (2018). Effect of Tb^{3+} doping on the photocatalytic activity of TiO_2 coatings formed by plasma electrolytic oxidation of titanium. *Surf. Coating. Technol.* 337: 279–289.
- 142 Stojadinović, S. and Ćirić, A. (2021). Photoluminescence of $\text{ZnO}:\text{Eu}^{3+}$ and $\text{ZnO}:\text{Tb}^{3+}$ coatings formed by plasma electrolytic oxidation of pure zinc substrate. *J. Lumin.* 235: 118022.

- 143 Stojadinović, S. and Vasilić, R. (2016). Orange-red photoluminescence of $\text{Nb}_2\text{O}_5:\text{Eu}^{3+}$, Sm^{3+} coatings formed by plasma electrolytic oxidation of niobium. *J. Alloys Compd.* 685: 881–889.
- 144 Stojadinović, S., Tadić, N., Ćirić, A., and Vasilić, R. (2018). Photoluminescence properties of Eu^{3+} doped HfO_2 coatings formed by plasma electrolytic oxidation of hafnium. *Opt. Mater.* 77: 19–24.
- 145 Ćirić, A. and Stojadinović, S. (2020). Structural and photoluminescence properties of Y_2O_3 and $\text{Y}_2\text{O}_3:\text{Ln}^{3+}$ ($\text{Ln} = \text{Eu}, \text{Er}, \text{Ho}$) films synthesized by plasma electrolytic oxidation of yttrium substrate. *J. Lumin.* 217: 116762.
- 146 Ćirić, A. and Stojadinović, S. (2019). Photoluminescence of Gd_2O_3 and $\text{Gd}_2\text{O}_3:\text{Ln}^{3+}$ ($\text{Ln} = \text{Eu}, \text{Er}, \text{Ho}$) formed by plasma electrolytic oxidation of pure gadolinium substrate. *Opt. Mater.* 99: 109546.
- 147 Kaseem, M., Fatimah, S., Nashrah, N., and Ko, Y.G. (2021). Recent progress in surface modification of metals coated by plasma electrolytic oxidation: principle, structure, and performance. *Prog. Mater. Sci.* 117: 100735.
- 148 Li, J., Meng, S., Niu, J., and Lu, H. (2017). Electronic structures and optical properties of monoclinic ZrO_2 studied by first-principles local density approximation + U approach. *J. Adv. Ceram.* 6: 43–49.
- 149 Cabello, G., Lillo, L., Caro, C. et al. (2008). Structure and optical characterization of photochemically prepared ZrO_2 thin films doped with erbium and europium. *J. Non-Cryst. Solids* 354: 3919–3928.
- 150 Martínez-Olmos, R.C., Guzmán-Mendoza, J., Báez-Rodríguez, A. et al. (2015). Synthesis, characterization and luminescence studies in $\text{ZrO}_2:\text{Dy}^{3+}$ and $\text{ZrO}_2:\text{Dy}^{3+}$, Gd^{3+} films deposited by the Pyrosol method. *Opt. Mater.* 46: 168–174.
- 151 Lu, X., Mohedano, M., Blawert, C. et al. (2016). Plasma electrolytic oxidation coatings with particle additions – a review. *Surf. Coat. Technol.* 307: 1165–1182.
- 152 Stojadinović, S., Radić-Perić, J., Vasilić, R., and Perić, M. (2014). Spectroscopic investigation of direct current (DC) plasma electrolytic oxidation of zirconium in citric acid. *Appl. Spectrosc.* 68: 101–112.
- 153 Schmid, H. (1994). Multi-ferroicmagnetoelectrics. *Ferroelectrics* 162 (1): 317–338.
- 154 Fiebig, M. (2005). Revival of the magnetoelectric effect. *J. Phys. D: Appl. Phys.* 38 (8): R123–R152.
- 155 Ramesh, R. and Spaldin, N.A. (2009). Multiferroics: progress and prospects in thin films. *Nanosci. Technol.* 3: 20–28.
- 156 Hill, N.A. (2000). Why are there so few magnetic ferroelectrics? *J. Phys. Chem. B* 104 (29): 6694–6709.
- 157 Israel, C., Mathur, N.D., and Scott, J.F. (2008). A one-cent room-temperature magnetoelectric sensor. *Nat. Mater.* 7 (2): 93–94.
- 158 Veev, J.P., Duan, C.-G., Burton, J.D. et al. (2009). Magnetic tunnel junctions with ferroelectric barriers: prediction of four resistance states from first principles. *Nano Lett.* 9 (1): 427–432.
- 159 Jahns, R., Greve, H., Woltermann, E. et al. (2012). Sensitivity enhancement of magnetoelectric sensors through frequency-conversion. *Sens. Actuators A Phys.* 183: 16–21.

- 160 Jahns, R., Piorra, A., Lage, E. et al. (2013). Giant magnetoelectric effect in thin-film composites. *J. Am. Ceram. Soc.* 96 (6): 1673–1681.
- 161 Moyerman, S., Eckert, J.C., Borchers, J.A. et al. (2006). Magnetic structure variations during giant magnetoresistance training in spin valves with picoscale antiferromagnetic layers. *J. Appl. Phys.* 99 (8): 8–11.
- 162 Li, M., Cui, W., Yu, J. et al. (2015). Magnetic proximity effect and interlayer exchange coupling of ferromagnetic/topological insulator/ferromagnetic trilayer. *Phys. Rev. B Condens. Matter Mater. Phys.* 91 (1): 1–7.
- 163 Daughton, J., Brown, J., Chen, E. et al. (1994). Magnetic field sensors using GMR multilayer. *IEEE Trans. Magn.* 30 (6): 4608–4610.
- 164 Prinz, G.A. (1998). Magnetoelectronics. *Science* 282 (5394): 1660–1663.
- 165 Ravi, S. (2020). High Curie temperature and room temperature magnetoresistance in $\text{Pr}_2\text{FeCrO}_6$ material for spintronics applications. *Mater. Lett.* 278: 128448.
- 166 Dieny, B., Speriosu, V.S., Parkin, S.S.P. et al. (1991). Giant magnetoresistive in soft ferromagnetic multilayers. *Phys. Rev. B* 43 (1): 1297–1300.
- 167 Heim, D.E., Fontana, R.E., Tsang, C. et al. (1994). Design and operation of spin valve sensors. *IEEE Trans. Magn.* 30 (2): 316–321.
- 168 Freitas, P.P., Leal, J.L., Melo, L.V. et al. (1994). Spin-valve sensors exchange-biased by ultrathin TbCo films. *Appl. Phys. Lett.* 65 (4): 493–495.
- 169 Leal, L., Oliveira, N., Rodrigues, L. et al. (1994). Unshielded spin-valve sensors exchange-biased by thin TbCo layers. *IEEE Trans. Magn.* 30 (6): 3831–3833.
- 170 Tsang, C., Fontana, R.E., Lin, T. et al. (1994). Design, fabrication & testing of spin-valve read heads for high density recording. *IEEE Trans. Magn.* 30 (6): 3801–3806.
- 171 Kosterev, A.A., Tittel, F.K., Serebryakov, D.V. et al. (2005). Applications of quartz tuning forks in spectroscopic gas sensing. *Rev. Sci. Instrum.* 76 (4): 043105–1–043105–9.
- 172 Valkering, A., Mares, A., Untiedt, C. et al. (2005). A force sensor for atomic point contacts. *Rev. Sci. Instrum.* 76 (10): 103903–1–103903–5.
- 173 Lee, S., Yun, M., and Jeon, S. (2014). A tunable microresonator sensor based on a photocrosslinking polymer microwire. *App. Phys. Lett.* 104: 053506–1–053506–4.
- 174 Kim, W., Yun, M., Lee, S., and Jeon, S. (2016). Enhanced sensitivity of quartz tuning fork sensors using electrospun polymer wires. *RSC Adv.* 6: 31131–31134.

